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Ti insertion in the MTe_5 (M = Zr, Hf) structure type: $Hf_{0.78}Ti_{0.22}Te_5$

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Key indicators: single-crystal X-ray study; T = 290 K; mean σ (Te–Te) = 0.001 Å; disorder in main residue; R factor = 0.029; wR factor = 0.068; data-to-parameter ratio = 23.2.

The group 5 mixed-metal telluride, $Hf_{0.78}Ti_{0.22}Te_5$ (hafnium titanium pentatelluride), is isostructural with the binary phases $HfTe_5$ and $ZrTe_5$ and forms a layered structure extending parallel to (010). The layers are made up from chains of bicapped metal-centered trigonal prisms and zigzag Te chains. The metal site (site symmetry m2m) is occupied by statistically disordered Hf [78.1 (5)%] and Ti [21.9 (5)%]. In addition to the regular Te—Te pair [2.7448 (13) Å] forming the short base of the equilateral triangle of the trigonal prism, an intermediate Te···Te separation [2.9129 (9) Å] is also found. The classical charge balance of the compound can be described as $[M^{4+}][Te^{2-}][Te_2^{2-}][Te_2^{0}]$ (M = Hf, Ti). The individual metal content can vary in different crystals, apparently forming a random substitutional solid solution ($Hf_{1-x}Ti_x$)Te₅, with 0.15 $\leq x \leq 0.22$.

Related literature

For the synthesis and structure of HfTe₅ and ZrTe₅, see: Brattås & Kjekshus (1971); Furuseth *et al.* (1973, 1975). For properties of HfTe₅ and ZrTe₅, see: DiSalvo *et al.* (1981). For extensive Te \cdots Te interactions in metal tellurides, see: Pell & Ibers (1996); Mar & Ibers (1993).

Experimental

Crystal data

	a = 3.9595 (3) Å
$M_r = 787.76$	<i>b</i> = 14.4350 (13) Å
Orthorhombic, Cmcm	c = 13.7062 (9) Å

 $V = 783.39 (10) \text{ Å}^3$ Z = 4Mo $K\alpha$ radiation

Data collection

Rigaku R-AXIS RAPID diffractometer Absorption correction: multi-scan (NUMABS; Higashi, 2000) $T_{min} = 0.263, T_{max} = 1.000$

Refinement

 $R[F^2 > 2\sigma(F^2)] = 0.029$ $wR(F^2) = 0.068$ S = 1.16533 reflections

Table 1

Selected geometric parameters (Å, °).

M = Hf or Ti

$\frac{M - \text{Te2}^{i}}{M - \text{Te1}^{ii}}$	2.9251 (6) 2.9446 (8)	M-Te3	2.9498 (7)
Te3 ⁱⁱ -Te3-Te3 ⁱⁱⁱ	85.63 (3)		

Symmetry codes: (i) $x + \frac{1}{2}$, $y + \frac{1}{2}$, z; (ii) $-x + \frac{3}{2}$, $-y + \frac{1}{2}$, -z; (iii) $-x + \frac{5}{2}$, $-y + \frac{1}{2}$, -z.

Data collection: *RAPID-AUTO* (Rigaku, 2006); cell refinement: *RAPID-AUTO*; data reduction: *RAPID-AUTO*; program(s) used to solve structure: *SHELXS97* (Sheldrick, 2008); program(s) used to refine structure: *SHELXL97* (Sheldrick, 2008); molecular graphics: *DIAMOND3* (Brandenburg, 1999); software used to prepare material for publication: *WinGX* (Farrugia, 1999).

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Supplementary data and figures for this paper are available from the IUCr electronic archives (Reference: WM2588).

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inorganic compounds

 $\mu = 28.76 \text{ mm}^{-1}$

 $0.30 \times 0.04 \times 0.02$ mm

3405 measured reflections

533 independent reflections

516 reflections with $I > 2\sigma(I)$

T = 290 K

 $R_{\rm int} = 0.059$

23 parameters $\Delta \rho_{\rm max} = 2.38 \text{ e} \text{ Å}^{-3}$

 $\Delta \rho_{\rm min} = -2.07 \ {\rm e} \ {\rm \AA}^{-3}$

supplementary materials

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Ti insertion in the MTe_5 (M = Zr, Hf) structure type: Hf_{0.78}Ti_{0.22}Te₅

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Comment

The existence of the binary pentatellurides, MTe_5 (M = Zr, Hf) was established in 1971 by Brattås & Kjekshus. These compounds have been the most chalcogen-rich stoichiometric transition metal binary chalcogenides since then. These phases are important not only because of their unusually high chalcogen contents but because of the anomalies in their transport properties (DiSalvo *et al.*, 1981). The Ti analogue of this phase is desired to study the relationship between structure and properties. However, efforts to find TiTe₅ have not been successful despite numerous attempts (Furuseth *et al.*, 1973). Indeed, no Ti analogue of the MTe_8 polyhedra found in MTe_5 (Furuseth *et al.*, 1975) has been observed up to now. During attempts to synthesize new metal tellurides, we found the new Ti-containing mixed-metallic phase Hf_{0.78}Ti_{0.22}Te₅.

The title compound is isostructural with HfTe₅ and ZrTe₅. Detailed descriptions of this structural type have been reported previously (Furuseth *et al.*, 1973). A view of the structure down the *a* axis is given in Fig. 1, which shows the layered nature parallel to (010). The structure is composed of chains made up from bicapped trigonal prismatic *M*Te₈ units. The metal (*M*) site occupied by statistically disordered Hf (78.1 (5) %) and Ti (21.9 (5) %) is surrounded by eight Te atoms. Three crystallographically independent Te sites are found in the title compound. Both Te1 (site symmetry *m*2*m*) and Te2 (site symmetry *m*..) are at the corners of a triangular prism and they are bridging the *M* atoms to form a chain. Two Te3 (site symmetry *m*..) atoms outside the rectangular faces of the prism are connected to the neighboring Te3 to form the infinite zigzag Te chain. Finally, the *M*Te₅ layer is formed by the alternate linking of these chains (Fig. 2).

The structure shows a wide range of Te···Te interactions. In the prism, the Te2—Te2 pair (2.7448 (13) Å) forming the short base of the equilateral triangle exhibits a regular Te—Te bond, Te_2^{2-} (*e.g.* discussed by Pell & Ibers, 1996). In addition, an intermediate Te3···Te3 separation (2.9129 (9) Å) is indicative of a weak single bond (Mar & Ibers, 1993) but we assign an oxidation state of 0 for Te3 ($^1_{\infty}$ [Te⁰]). The classical charge balance of the compound can be described as $[M^{4+}]$ [Te²⁻][Te₂²⁻][Te₂⁰] (M=Hf, Ti).

Structury analysis of three different crystals from the same reaction tube showed that the metal content can vary, apparently forming a random substitutional solid solution $(Hf_{1-x}Ti_x)Te_5, 0.15 \le x \le 0.22.$

Experimental

The title compound, $Hf_{0.78}Ti_{0.22}Te_5$, was prepared by the reaction of the elements with the use of the reactive halide-flux technique. Hf powder (CERAC 99.8%), Ti powder (CERAC 99.5%), and Te powder (CERAC 99.95%) were mixed in a fused silica tube in a molar ratio of Hf: Ti: Te = 1: 1: 10 and then CsCl (CERAC 99.9%) was added in a weight ratio of HfTiTe₁₀: CsCl = 1: 2. The tube was evacuated to 0.133 Pa, sealed and heated gradually (50 K/h) to 650 K, where it was kept for 72 h. The tube was then cooled to room temperature at the rate of 3 K/h. The excess halide was removed with distilled water and dark block-shaped crystals were obtained. The crystals are stable in air and water. A qualitative X-ray fluorescence analysis of the crystals indicated the presence of Hf, Ti, and Te. A quantitative XRF analysis indicated that

the Hf: Ti ratio is 80: 20. The composition of the compound was determined by single-crystal X-ray diffraction.

Refinement

The statistically disordered nature of the *M* site in the title compound was checked by refining the anisotropic displacement parameters (ADPs). When the model was refined assuming HfTe₅ or TiTe₅, the displacement parameters of the metal site were very large and small, respectively. In both cases the reliability indices were rather high (wR2 > 0.127). In the refined mixed-metal model, the ADPs of the metal atoms are comparable with those of the other atoms and the residuals were reduced significantly (wR2 = 0.068). The remaining highest and lowest electron densities are found 0.81 and 0.77 Å from atom Te1.

Computing details

Data collection: *RAPID-AUTO* (Rigaku, 2006); cell refinement: *RAPID-AUTO* (Rigaku, 2006); data reduction: *RAPID-AUTO* (Rigaku, 2006); program(s) used to solve structure: *SHELXS97* (Sheldrick, 2008); program(s) used to refine structure: *SHELXL97* (Sheldrick, 2008); molecular graphics: *DIAMOND3* (Brandenburg, 1999); software used to prepare material for publication: *WinGX* (Farrugia, 1999).

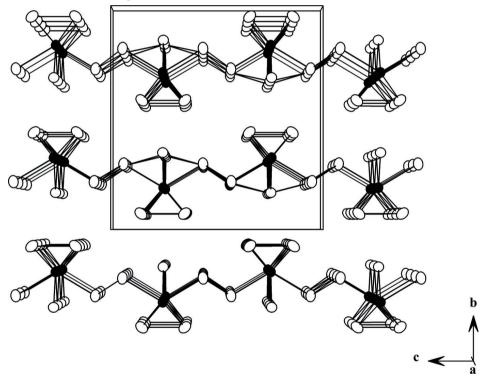


Figure 1

A perspective view of $Hf_{0.78}Ti_{0.22}Te_5$ down the *a* axis showing the stacking of the layers. Filled and open circles represent metals and S atoms, respectively. The displacement ellipsoids are drawn at the 90% probability level.

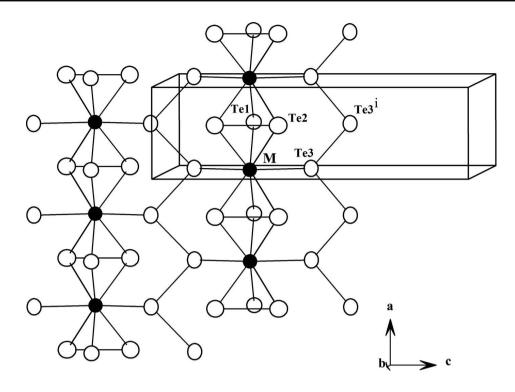


Figure 2

A view of the layer of $Hf_{0.78}Ti_{0.22}Te_5$. Anisotropic displacement ellipsoids are drawn at the 90% probability level. Atoms are as marked as in Fig. 1. [Symmetry code: (i) 1/2 + x, 1/2 - y, 1 - z].

F(000) = 1284

 $\theta = 3.2 - 27.5^{\circ}$ $\mu = 28.76 \text{ mm}^{-1}$

Needle, black

 $0.30 \times 0.04 \times 0.02 \text{ mm}$

T = 290 K

 $D_{\rm x} = 6.679 {\rm Mg} {\rm m}^{-3}$

Mo K α radiation, $\lambda = 0.71073$ Å

Cell parameters from 3264 reflections

Hafnium titanium pentatelluride

Crystal data

Hf_{0.78}Ti_{0.22}Te₅ $M_r = 787.76$ Orthorhombic, *Cmcm* Hall symbol: -C 2c 2 a = 3.9595 (3) Å b = 14.4350 (13) Å c = 13.7062 (9) Å V = 783.39 (10) Å³ Z = 4

Data collection

Rigaku R-AXIS RAPID	3405 measured reflections
diffractometer	533 independent reflection
Radiation source: fine-focus sealed tube	516 reflections with $I > 2\sigma$
Graphite monochromator	$R_{\rm int} = 0.059$
ω scans	$\theta_{\text{max}} = 27.5^{\circ}, \ \theta_{\text{min}} = 2.8^{\circ}$
Absorption correction: multi-scan	$h = -4 \rightarrow 5$
(NUMABS; Higashi, 2000)	$k = -18 \rightarrow 18$
$T_{\min} = 0.263, \ T_{\max} = 1.000$	$l = -17 \rightarrow 17$

flections flections h $I > 2\sigma(I)$ 2.8° Refinement

Refinement on F^2	Secondary atom site location: difference Fourier
Least-squares matrix: full	map
$R[F^2 > 2\sigma(F^2)] = 0.029$	$w = 1/[\sigma^2(F_o^2) + (0.0304P)^2 + 7.5394P]$
$wR(F^2) = 0.068$	where $P = (F_0^2 + 2F_c^2)/3$
S = 1.16	$(\Delta/\sigma)_{\rm max} < 0.001$
533 reflections	$\Delta ho_{ m max} = 2.38 \ { m e} \ { m \AA}^{-3}$
23 parameters	$\Delta ho_{ m min}$ = -2.07 e Å ⁻³
0 restraints	Extinction correction: SHELXL97 (Sheldrick,
Primary atom site location: structure-invariant	2008)
direct methods	Extinction coefficient: 0.00109 (14)

Special details

Geometry. All s.u.'s (except the s.u. in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell s.u.'s are taken into account individually in the estimation of s.u.'s in distances, angles and torsion angles; correlations between s.u.'s in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell s.u.'s is used for estimating s.u.'s involving l.s. planes.

Refinement. Refinement of F^2 against ALL reflections. The weighted *R*-factor *wR* and goodness of fit *S* are based on F^2 , conventional *R*-factors *R* are based on *F*, with *F* set to zero for negative F^2 . The threshold expression of $F^2 > 2\sigma(F^2)$ is used only for calculating *R*-factors(gt) *etc.* and is not relevant to the choice of reflections for refinement. *R*-factors based on F^2 are statistically about twice as large as those based on *F*, and *R*-factors based on ALL data will be even larger.

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters $(Å^2)$

	x	У	Ζ	$U_{ m iso}$ */ $U_{ m eq}$	Occ. (<1)
Hf	1	0.31435 (5)	0.25	0.0109 (3)	0.781 (5)
Ti	1	0.31435 (5)	0.25	0.0109 (3)	0.219 (5)
Te1	1	0.33665 (6)	-0.25	0.0128 (3)	
Te2	1	-0.07070 (5)	0.14987 (5)	0.0169 (3)	
Te3	1	0.20952 (5)	0.06526 (4)	0.0154 (2)	

	U^{11}	U^{22}	U^{33}	U^{12}	U^{13}	U^{23}
Hf	0.0103 (4)	0.0122 (4)	0.0102 (4)	0	0	0
Ti	0.0103 (4)	0.0122 (4)	0.0102 (4)	0	0	0
Te1	0.0129 (4)	0.0121 (5)	0.0134 (4)	0	0	0
Te2	0.0164 (4)	0.0164 (4)	0.0178 (4)	0	0	0.0051 (2)
Te3	0.0155 (4)	0.0190 (4)	0.0116 (3)	0	0	-0.0010 (2)

Atomic displacement parameters $(Å^2)$

Geometric parameters (Å, °)

M—Te2 ⁱ	2.9251 (6)	Te1—M ^v	2.9446 (8)
M—Te2 ⁱⁱ	2.9251 (6)	Te1—M ^{vi}	2.9446 (8)
M—Te2 ⁱⁱⁱ	2.9251 (6)	Te1—M ^{vi}	2.9446 (8)
M—Te2 ^{iv}	2.9251 (6)	Te2—Te2 ^{vii}	2.7448 (13)
M—Te1 ^v	2.9446 (8)	Te2—M ^{viii}	2.9251 (6)
M—Te1 ^{vi}	2.9446 (8)	Te2—M ^{viii}	2.9251 (6)
M—Te3	2.9498 (7)	Te2—M ^{ix}	2.9251 (6)
M—Te3 ^{vii}	2.9498 (7)	Te2—M ^{ix}	2.9251 (6)
Te1—M ^v	2.9446 (8)	Te3—Te3 ^v	2.9129 (9)

Te2 ⁱ —M—Te2 ⁱⁱ	55.96 (3)	Te2 ^{iv} —M—Te3 ^{vii}	83.583 (16)
Te2 ⁱ —M—Te2 ⁱⁱⁱ	110.88 (3)	Te1 ^v —M—Te3 ^{vii}	67.683 (15)
Te2 ⁱⁱ —M—Te2 ⁱⁱⁱ	85.19 (2)	Te1 ^{vi} —M—Te3 ^{vii}	67.683 (15)
Te2 ⁱ —M—Te2 ^{iv}	85.19 (2)	Te3—M—Te3 ^{vii}	118.27 (3)
Te2 ⁱⁱ —M—Te2 ^{iv}	110.88 (3)	M^v —Te1— M^v	0.000 (18)
Te2 ⁱⁱⁱ —M—Te2 ^{iv}	55.96 (3)	M^v —Te1— M^{vi}	84.50 (3)
Te2 ⁱ —M—Te1 ^v	151.040 (16)	M^{v} —Te1— M^{vi}	84.50 (3)
Te2 ⁱⁱ —M—Te1 ^v	151.040 (16)	M^{v} —Te1— M^{vi}	84.50 (3)
Te2 ⁱⁱⁱ —M—Te1 ^v	87.986 (16)	M^{v} —Te1— M^{vi}	84.50 (3)
$Te2^{iv}$ —M— $Te1^{v}$	87.986 (16)	M^{vi} —Te1— M^{vi}	0
$Te2^{i}$ —M— $Te1^{vi}$	87.986 (16)	Te2 ^{vii} —Te2—M ^{viii}	62.019 (13)
Te2 ⁱⁱ —M—Te1 ^{vi}	87.986 (16)	Te2 ^{vii} —Te2—M ^{viii}	62.019 (13)
Te2 ⁱⁱⁱ —M—Te1 ^{vi}	151.040 (16)	M ^{viii} —Te2—M ^{viii}	0.00(2)
$Te2^{iv}$ —M— $Te1^{vi}$	151.040 (16)	Te2 ^{vii} —Te2—M ^{ix}	62.019 (13)
Te1 ^v —M—Te1 ^{vi}	84.49 (3)	M^{viii} —Te2— M^{ix}	85.19 (2)
Te2 ⁱ —M—Te3	133.927 (10)	M ^{viii} —Te2—M ^{ix}	85.19 (2)
Te2 ⁱⁱ —M—Te3	83.584 (16)	Te2 ^{vii} —Te2—M ^{ix}	62.019 (13)
Te2 ⁱⁱⁱ —M—Te3	83.583 (16)	M^{viii} —Te2— M^{ix}	85.19 (2)
Te2 ^{iv} —M—Te3	133.927 (10)	M^{viii} —Te2— M^{ix}	85.19 (2)
Te1 ^v —M—Te3	67.683 (15)	M ^{ix} —Te2—M ^{ix}	0.00(2)
Te1 ^{vi} —M—Te3	67.683 (15)	Te3 ^v —Te3—Te3 ^{vi}	85.63 (3)
Te2 ⁱ —M—Te3 ^{vii}	83.583 (16)	Te3 ^v —Te3—M	108.74 (3)
Te2 ⁱⁱ —M—Te3 ^{vii}	133.927 (10)	Te3 ^{vi} —Te3—M	108.74 (3)
Te2 ⁱⁱⁱ —M—Te3 ^{vii}	133.927 (10)		

Symmetry codes: (i) *x*+1/2, *y*+1/2, *-z*+1/2; (ii) *x*+1/2, *y*+1/2, *z*; (iii) *x*-1/2, *y*+1/2, *z*; (iv) *x*-1/2, *y*+1/2, *-z*+1/2; (v) *-x*+3/2, *-y*+1/2, *-z*; (vi) *-x*+5/2, *-y*+1/2, *-z*; (vii) *x*, *y*, *-z*+1/2; (viii) *x*-1/2, *y*-1/2, *z*; (ix) *x*+1/2, *y*-1/2, *z*.